

Acta Cryst. (1995), C51, 1731–1732

Paraelastic $\text{Pb}_5\text{Al}_3\text{F}_{19}$ at 370 K

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(Received 24 October 1994; accepted 8 February 1995)

Abstract

The paraelastic phase of lead aluminium fluoride, $\text{Pb}_5\text{Al}_3\text{F}_{19}$, forms above a transition temperature of about 360 K. The structure contains corner-sharing chains of AlF_6 octahedra, individual AlF_6 octahedra, individual F^- ions and Pb^{2+} ions. Both sets of independent AlF_6 octahedra are slightly distorted, with $\text{Al}-\text{F}$ distances of 1.79 (3) and 1.80 (2) Å. The transition from the ferroelastic phase is weakly first order [Ravez, Andriamampianina, Simon, Rabardel, Ihringer & Abrahams (1994). *J. Appl. Cryst.* **27**, 362–368].

Comment

$\text{Pb}_5\text{Al}_3\text{F}_{19}$ has been reported to undergo three phase transitions on cooling from 400 to 80 K (Ravez, Andriamampianina, Simon, Rabardel, Ihringer & Abrahams, 1994). The structure of the antiferroelectric phase with space group $P4/n$ at room temperature, together with structural changes that occur at the transition from the ferroelectric phase with space group $I4cm$, have been described previously (Andriamampianina, Gravereau, Ravez & Abrahams, 1994).

The phase-transition sequence in $\text{Pb}_5\text{Al}_3\text{F}_{19}$ is the following (with transition temperatures, on cooling):

ferroelectric $\xleftarrow{120 \text{ K}}$ antiferroelectric $\xleftarrow{305 \text{ K}}$ ferroelastic $\xleftarrow{360 \text{ K}}$ paraelastic

X-ray diffraction measurements have been made on a crystal, prepared by Ravez, Andriamampianina & Abrahams (1994), in the paraelastic phase at 370 K, using a heated flow of dry nitrogen gas. Absorption correction error was minimized by grinding the crystal to a sphere. The symmetry of the intensity distribution

and the diffraction conditions $h+k+l=2n$ lead to a choice of space groups $I4$, $\bar{I}\bar{4}$ or $I4/m$.

Neither the pyroelectric p_3 nor the piezoelectric d_{33} coefficients have measurable magnitudes, suggesting that the point group is centrosymmetric. The two independent Al atoms occupy AlF_6 octahedra and the Pb atoms occupy tunnels of 2.5 Å radii.

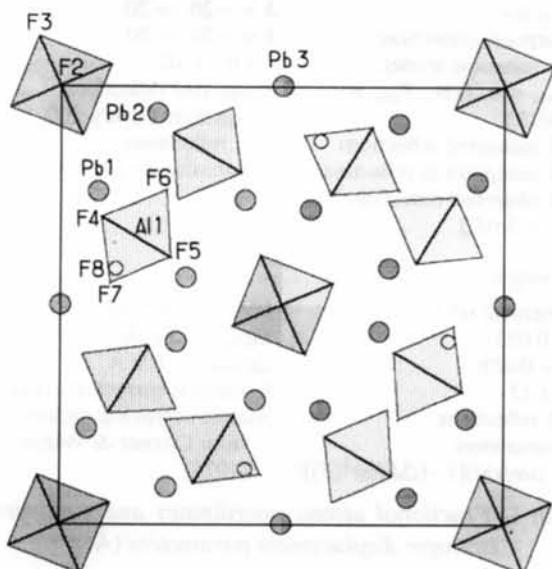


Fig. 1. Projection of the paraelastic phase of $\text{Pb}_5\text{Al}_3\text{F}_{19}$ along the c axis, with AlF_6 octahedra shown shaded. The larger filled circles represent Pb and the smaller open circles F8 atoms (see Table 1).

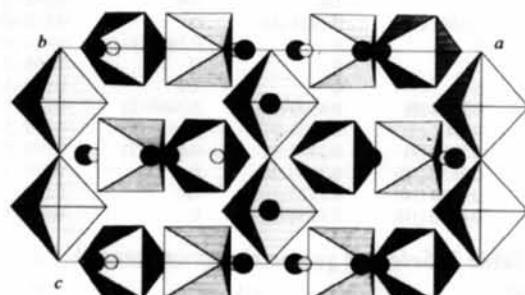


Fig. 2. Projection of the paraelastic phase of $\text{Pb}_5\text{Al}_3\text{F}_{19}$ along the b axis, with AlF_6 octahedra shown shaded. The larger filled circles represent Pb and the smaller open circles F8 atoms (see Table 1).

Experimental

The crystal was prepared by melting and slow cooling (Ravez, Andriamampianina & Abrahams, 1994). The density D_m was measured by flotation.

Crystal data

$\text{Pb}_5\text{Al}_3\text{F}_{19}$	Mo $K\alpha$ radiation
$M_r = 1477.89$	$\lambda = 0.71073 \text{ \AA}$
Tetragonal	Cell parameters from 25 reflections
$I4/m$	$\theta = 10^\circ - 26^\circ$
$a = 14.285 (7) \text{ \AA}$	$\mu = 56.4 \text{ mm}^{-1}$
$c = 7.227 (3) \text{ \AA}$	

$V = 1475 (3) \text{ \AA}^3$
 $Z = 4$
 $D_x = 6.66 \text{ Mg m}^{-3}$
 $D_m = 6.66 \text{ Mg m}^{-3}$

$T = 370 \text{ K}$
Sphere
0.3 mm (radius)
Colourless

Data collection

Enraf–Nonius CAD-4
diffractometer
 ω/θ scans
Absorption correction:
by spherical model
 $T_{\min} = 0.0009$, $T_{\max} = 0.0185$
5787 measured reflections
1715 independent reflections
1314 observed reflections
 $[I > 3\sigma(I)]$

$R_{\text{int}} = 0.065$
 $\theta_{\text{max}} = 30^\circ$
 $h = -20 \rightarrow 20$
 $k = -20 \rightarrow 20$
 $l = 0 \rightarrow 10$
3 standard reflections
monitored every 200
reflections
intensity decay: <4%

Refinement

Refinement on F
 $R = 0.053$
 $wR = 0.058$
 $S = 1.12$
1115 reflections
52 parameters
 $w = (\text{weight})[1 - (\Delta F/6\sigma^2(F))^2]$ (1974)

$(\Delta/\sigma)_{\text{max}} = 0.01$
 $\Delta\rho_{\text{max}} = 5 \text{ e \AA}^{-3}$
 $\Delta\rho_{\text{min}} = -6 \text{ e \AA}^{-3}$
Extinction correction: none
Atomic scattering factors
from Cromer & Waber
(1974)

Table 1. Fractional atomic coordinates and equivalent isotropic displacement parameters (\AA^2)

$$U_{\text{eq}} = (1/3)\sum_i \sum_j U_{ij} a_i^* a_j^* \mathbf{a}_i \cdot \mathbf{a}_j.$$

	x	y	z	U_{eq}
Pb1	0.08474 (7)	-0.23970 (6)	0	0.018 (1)
Pb2	-0.06116 (6)	-0.21768 (5)	1/2	0.013 (2)
Pb3	0	1/2	1/4	0.072 (2)
Al1	0.1581 (4)	0.3255 (4)	0	0.012 (4)
Al2	0	0	0.2460 (1)	0.012 (4)
F1	0	0	0	0.038 (5)
F2	0	0	1/2	0.038 (5)
F3	0.3854 (9)	0.4500 (8)	0.2460 (2)	0.038 (5)
F4	0.2001 (9)	0.4076 (9)	0.1730 (2)	0.038 (5)
F5	0.3921 (9)	0.2494 (9)	0.6700 (2)	0.038 (5)
F6	0.2630 (10)	0.2610 (10)	0	0.038 (5)
F7	0.4530 (10)	0.1120 (10)	1/2	0.038 (5)
F8	0.4170 (10)	0.1250 (10)	0	0.038 (5)

Table 2. Selected geometric parameters (\AA , $^\circ$)

Pb1...F4 ⁱ	2.52 (1)	Al1—F4 ^{xi}	1.82 (1)
Pb1...F4 ⁱⁱ	2.52 (1)	Al1—F5 ^{vii}	1.78 (1)
Pb1...F6 ⁱⁱⁱ	2.54 (2)	Al1—F5 ^{xii}	1.78 (1)
Pb2...F5 ^{iv}	2.52 (1)	Al1—F6	1.75 (2)
Pb2...F5 ^v	2.52 (1)	Al1—F7 ^{vii}	1.81 (2)
Pb2...F6 ^{vi}	2.54 (2)	Al2—F1	1.777 (8)
Pb2...F8 ^{vi}	2.27 (2)	Al2—F2	1.836 (8)
Pb3...F7 ^{vii}	2.51 (1)	Al2—F3 ^{vii}	1.79 (1)
Pb3...F7 ^{viii}	2.51 (1)	Al2—F3 ^{xiii}	1.79 (1)
Pb3...F7 ^{ix}	2.51 (1)	Al2—F3 ^{xiv}	1.79 (1)
Pb3...F7 ^x	2.51 (1)	Al2—F3 ⁱ	1.79 (1)
Al1—F4	1.82 (1)		
F4 ⁱ ...Pb1...F4 ⁱⁱ	139.7 (6)	F4—Al1—F5 ^{xii}	92.7 (6)
F4 ⁱⁱ ...Pb1...F6 ⁱⁱⁱ	85.0 (3)	F5 ^{xii} —Al1—F5 ^{vii}	87.2 (9)
F5 ^v ...Pb2...F5 ^{iv}	142.2 (6)	F4—Al1—F6	93.3 (6)
F5 ^v ...Pb2...F6 ^{vi}	73.4 (3)	F4—Al1—F7 ^{vii}	88.4 (6)
F5 ^v ...Pb2...F8 ^{vi}	77.2 (6)	F5 ^{vii} —Al1—F7 ^{vii}	86.8 (6)
F6 ^{vi} ...Pb2...F8 ^{vi}	75.2 (6)	F6—Al1—F7 ^{xii}	177.7 (9)
F7 ^{vii} ...Pb3...F7 ^{viii}	87.9 (6)	F1—Al2—F2	179.99
F7 ^{ix} ...Pb3...F7 ^{viii}	121.2 (3)	F1—Al2—F3 ^{vii}	92.0 (5)
F4—Al1 ^{xi} —F4 ^{xi}	87.0 (8)	F3 ^{xiii} —Al2—F2 ^{vii}	89.93 (3)
F4—Al1—F5 ^{vii}	175.1 (7)	F3 ^{xiii} —Al2—F3 ^{vii}	176.1 (10)

Symmetry codes: (i) $\frac{1}{2} - y, x - \frac{1}{2}, \frac{1}{2} - z$; (ii) $\frac{1}{2} - y, x - \frac{1}{2}, z - \frac{1}{2}$; (iii) $y, -x, -z$; (iv) $x - \frac{1}{2}, y - \frac{1}{2}, \frac{3}{2} - z$; (v) $x - \frac{1}{2}, y - \frac{1}{2}, z - \frac{1}{2}$; (vi) $x - \frac{1}{2}, y - \frac{1}{2}, \frac{1}{2} + z$; (vii) $\frac{1}{2} - x, \frac{1}{2} - y, \frac{1}{2} - z$; (viii) $x - \frac{1}{2}, \frac{1}{2} + y, \frac{1}{2} - z$; (ix) $-y, x, z$; (x) $y, 1 - x, 1 - z$; (xi) $x, y, -z$; (xii) $\frac{1}{2} - x, \frac{1}{2} - y, z - \frac{1}{2}$; (xiii) $x - \frac{1}{2}, y - \frac{1}{2}, \frac{1}{2} - z$; (xiv) $y - \frac{1}{2}, \frac{1}{2} - x, \frac{1}{2} - z$.

Data collection: *CAD-4 Software* (Enraf–Nonius, 1989). Cell refinement: *CAD-4 Software*. Data reduction: local program (unpublished). Program(s) used to refine structure: *CRYSTALS* (Watkin, Carruthers & Betteridge, 1985).

The authors wish to thank R. Von der Mühl for piezoelectric and pyroelectric tests and D. J. Watkin for providing the refinement program.

Lists of structure factors, anisotropic displacement parameters and complete geometry have been deposited with the IUCr (Reference: DU1109). Copies may be obtained through The Managing Editor, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

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Acta Cryst. (1995). **C51**, 1732–1735

Octachlorosilasesquioxane, $\text{Cl}_8\text{Si}_8\text{O}_{12}$

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(Received 18 December 1994; accepted 27 March 1995)

Abstract

The structure of $\text{Cl}_8\text{Si}_8\text{O}_{12}$ is compared with those of the isostructural compounds $\text{H}_8\text{Si}_8\text{O}_{12}$ and $(\text{CH}_3)_8\text{Si}_8\text{O}_{12}$. The geometric deformations in $\text{Cl}_8\text{Si}_8\text{O}_{12}$ seem to